

SYMULACYJNE MODELOWANIE PROCESÓW DYFUZYJNYCH



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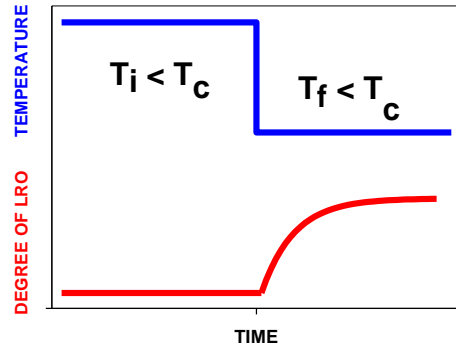
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OUTLINE:

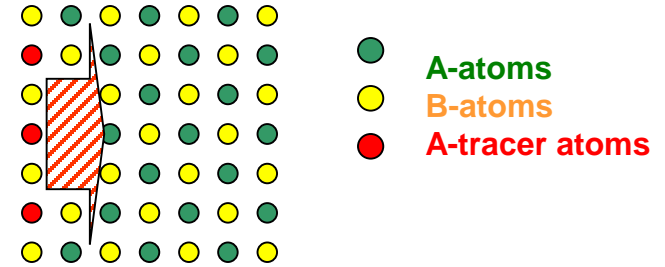
- ATOMIC-MIGRATION-CONTROLLED PHENOMENA IN SOLIDS
- EXPERIMENTAL RESULTS on self-diffusion and „order-order” kinetics in intermetallics: **thermodynamic activation energies** $E_A^{(D)}$ and $E_A^{(o-o)}$
 - fcc intermetallics (**Ni₃Al**)
 - bcc intermetallics (**NiAl**)
- MONTE CARLO ATOMISTIC SIMULATION of self-diffusion and „order-order” kinetics:
 - Methodology: (i) model, (ii) **SGCMC**: vacancy thermodynamics, (iii) **KMC**: vacancy-mediated atomic migration.
 - Results: **thermodynamic activation energies**
- ORIGIN OF $E_A^{(D)} > E_A^{(o-o)}$: elucidation in terms of thermal activation of elementary atomic jumps

„ORDER-ORDER”

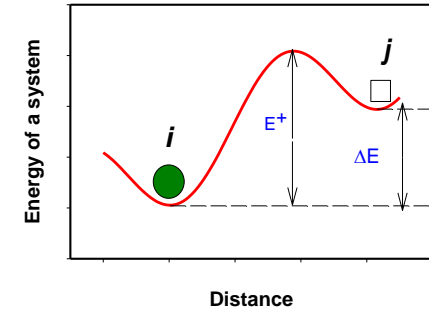
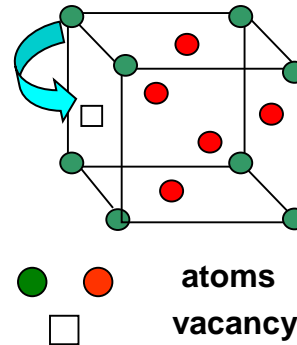


AND in intermetallics

DIFFUSION



COMMON ELEMENTARY MECHANISM:
atomic jumps to vacancies



SPECIFIC FEATURES: **correlation of atomic jumps**

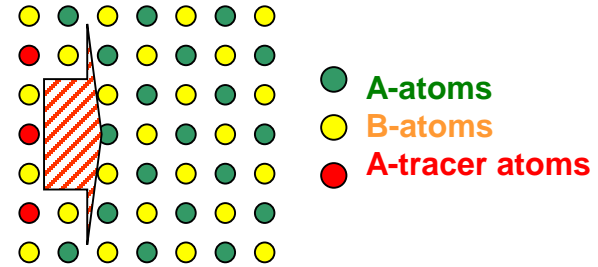
Minimisation of the energetic cost of local LRO perturbation by jumping atoms:
„six-jump-cycle”, ASB, triple defect mechanism, antisite diffusion etc.

Formation of equilibrium atomic configuration:
generation/elimination of antisite defects

CONSEQUENCE: COMPLEMENTARY INSIGHT INTO ATOMIC JUMP DYNAMICS

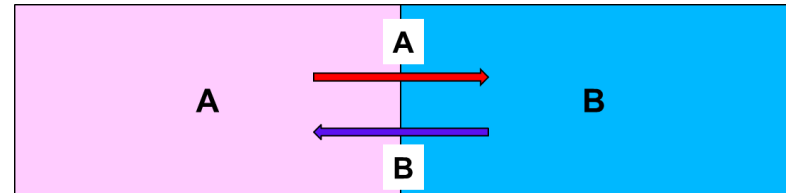
TYPES OF DIFFUSION

Self- (tracer) diffusion



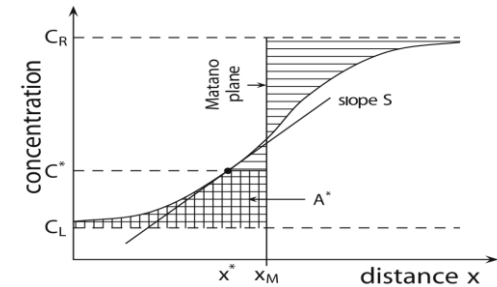
$$J_A = -D_A \times \frac{\delta C_A}{\delta x} \quad D_A = \lim_{t \rightarrow \infty} \left[\frac{1}{6} \frac{\partial}{\partial t} (\langle R_A^2(t) \rangle) \right]$$

Interdiffusion



$$J_{A(B)} = -\tilde{D} \times \frac{\delta C_{A(B)}}{\delta x}$$

\tilde{D} may be evaluated by analysing the concentration profile in the diffusion couple (**Boltzmann-Matano analysis**)



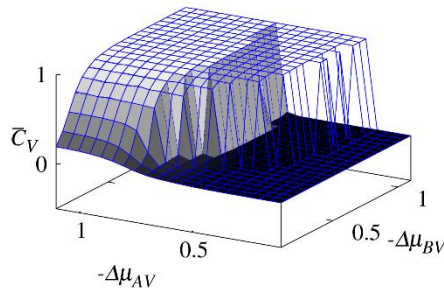
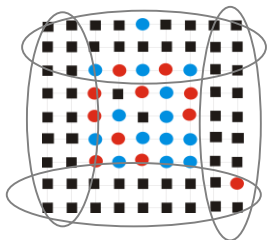
Any process controlled by (vacancy-mediated) atomic migration may be modelled by means of atomistic simulations:

- Monte Carlo – long time-scales covered
- Molecular Dynamics – only short time-scales covered

Semi Grand Canonical Monte Carlo (SGCMC):

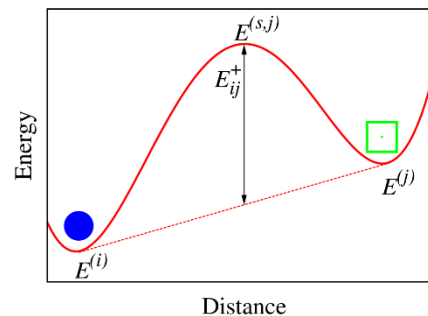
Fast determination of equilibrium parameters (atomic configuration, defect (vacancy) concentration)

$$\Pi_{p \rightarrow q} = \min \left\{ 1, \exp \left[- \frac{\Delta E_{p \rightarrow q} - (\Delta \mu_{qV} - \Delta \mu_{pV})}{k_B T} \right] \right\}$$



Kinetic Monte Carlo (KMC):

Combined with SGCMC enables simulation of steady-state atomic migration (at equilibrium configuration and vacancy concentration) with particular (e.g. vacancy) mechanism



$$P_{i \rightarrow j} = \frac{\tau \times \exp \left[- \frac{E_{ij}^{(s)}}{k_B T} \right]}{\sum_{ij} \exp \left[- \frac{E_{ij}^{(s)}}{k_B T} \right]}$$

$$\Delta t = \tau \times \left(\sum_{ij} \exp \left[- \frac{E_{ij}^{(s)}}{k_B T} \right] \right)^{-1}$$

DIRECT KMC SIMULATION RESULT:

$w_{\mu \rightarrow \nu}^{(X)}$ ← frequency of X-atom jumps from μ -sublattice to vacancies on ν -sublattice

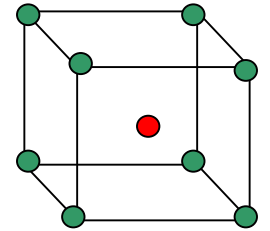
Interpretation:

$$w_{\alpha \rightarrow \beta}^{(A)}(t, T) = 8 \times \Pi_0 \times N_A \times C_A^{(\alpha)}(t, T) \times p_{AV}^{(\alpha\beta)}(t, T) \times \exp \left[- \frac{\langle E_{A,i(\alpha) \rightarrow j(\beta)}^{(m)}(t, T) \rangle}{k_B T} \right]$$

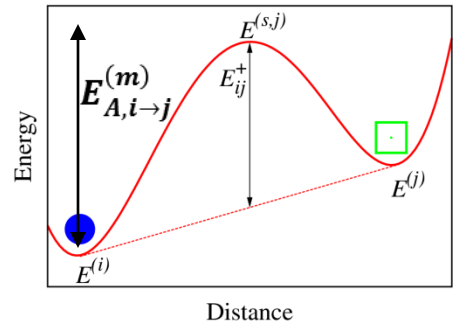
$T < T_C$

$$C_A^{(\beta)}(T, t) = [C_A^{(\beta)}]_0 \times \exp \left[- \frac{E_{F,A}^{(ant)}(t)}{k_B T} \right]$$

$$p_{AV}^{(\beta\alpha)}(t, T) \times \exp \left[- \frac{E_{A,i(\beta) \rightarrow j(\alpha)}^{(m)}(t)}{k_B T} \right] \sim \exp \left[- \frac{E_A^{(m,eff)}}{k_B T} \right]$$



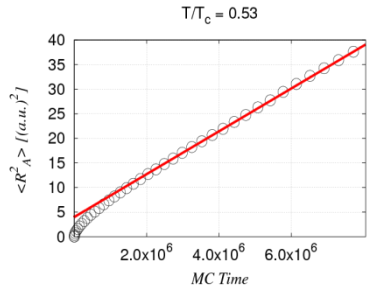
● α sublattice
● β sublattice



Self-diffusion

Measured at $C_{A(B)}^{(\beta(\alpha))} = \text{const}$

$$w_{\alpha \rightarrow \beta}^{(A)}(T) = w_{\beta \rightarrow \alpha}^{(A)}(T) = w_{eq}^{(A)}(T)$$



$$D_X = \lim_{t \rightarrow \infty} \left[\frac{1}{6} \frac{\partial}{\partial t} \left(\langle R_X^2(t) \rangle \right) \right]$$

$$D_A(T) = \frac{a^2}{3N_A} \times w_{eq}^{(A)}(T) \times f_A^{(corr)}(T)$$

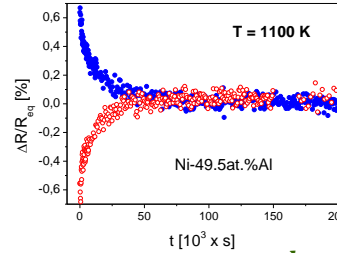
$$D_{A(B)}(T) = D_0 \times \exp \left[-\frac{E_A^{(D,A(B))}(T)}{k_B T} \right]$$

$$w_{\alpha \rightarrow \beta}^{(A)}(t, T) = [w_{\alpha \rightarrow \beta}^{(A)}]_0 \times \exp \left[-\frac{\langle (E_{A,i(\alpha) \rightarrow j(\beta)}^{(m)})(t, T) \rangle}{k_B T} \right]$$

$$f_{A(B)}^{(corr)}(T) = [f_{A(B)}^{(corr)}]_0 \times \exp \left[-\frac{E_{A(B)}^{(corr)}}{k_B T} \right]$$

$$E_A^{(D,A(B))} = \langle (E_{A,i(\alpha) \rightarrow j(\beta)}^{(m)})(t, T) \rangle + E_{A(B)}^{(corr)}$$

„Order-order” relaxation



Definition of the relaxation time τ

$$\frac{\eta(t, T) - \eta(t \rightarrow \infty, T)}{\Delta \eta_{tot}} = \exp \left[-\frac{t}{\tau(T)} \right]$$

$$\frac{d\eta}{dt}(t, T) = -\frac{1}{\tau(T)} \times [\eta(t, T) - \eta(t \rightarrow \infty, T)]$$

Atomistically:

$$\frac{dC_A^{(\beta)}}{dt}(t, T) = -\frac{d\eta_A}{2dt}(t, T) = \frac{1}{N_A} [w_{\alpha \rightarrow \beta}^{(A)}(t, T) - w_{\beta \rightarrow \alpha}^{(A)}(t, T)]$$

$$\frac{1}{\tau_A(T)} = \frac{1}{2N_A} \times \frac{w_{\alpha \rightarrow \beta}^{(A)}(t, T) - w_{\beta \rightarrow \alpha}^{(A)}(t, T)}{C_A^{(\beta)}(t \rightarrow \infty, T) - C_A^{(\beta)}(t, T)}$$

$$\tau(T)^{-1} = \tau_0^{-1} \times \exp \left[-\frac{E_A^{(O-O)}(T)}{k_B T} \right]$$

$$[w_{\alpha \rightarrow \beta}^{(A)}(t, T) - w_{\beta \rightarrow \alpha}^{(A)}(t, T)] \approx [\Delta w_{\alpha \rightarrow \beta}^{(A)}]_0 \times \exp \left[-\frac{E_{A(B)}^{(eff,dis)}}{k_B T} \right]$$

$$[C_{A(B)}^{(\beta(\alpha))}(t \rightarrow \infty, T) - C_{A(B)}^{(\beta(\alpha))}(t_0, T)] \approx [\Delta C_{A(B)}^{(\beta(\alpha))}]_0 \times \exp \left[-\frac{E_{F,A(B)}^{(ant)}}{k_B T} \right]$$

$$E_A^{(O-O,A(B))} = E_{A(B)}^{(eff,dis)} - E_{F,A(B)}^{(ant)}$$

Elucidable topics

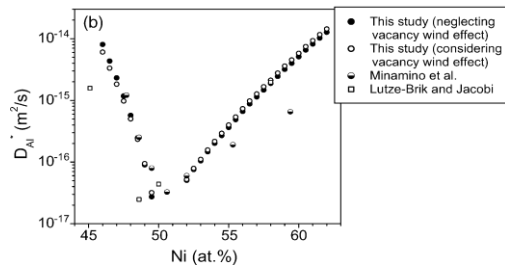
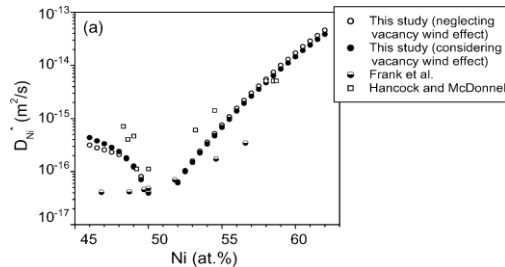
• Interrelation diffusion – relaxations

Example: $E_A^{(D)} > E_A^{(O-O)}$ in Ni-Al

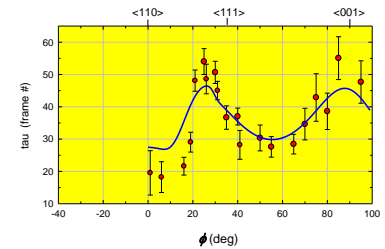
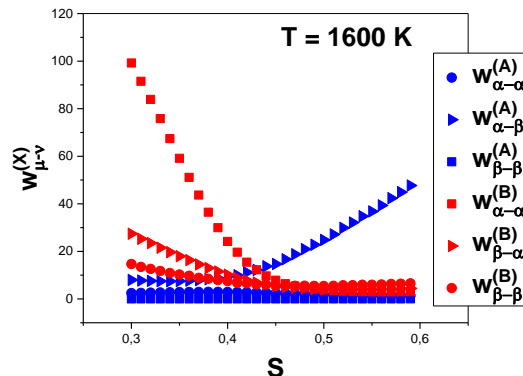
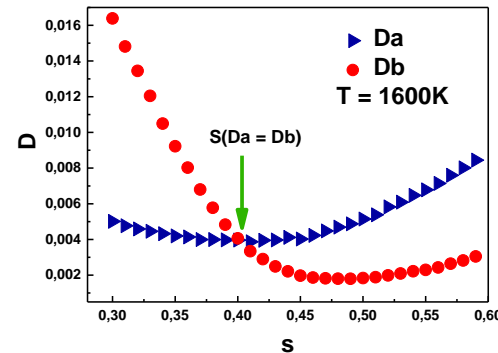
P.Sowa et al. Philos.Mag. 2017

• Features of self-diffusion

Example:



Ni-Al



Single atom jump determined with XPCS - (100) NNN jumps

THANK YOU FOR YOUR
ATTENTION

